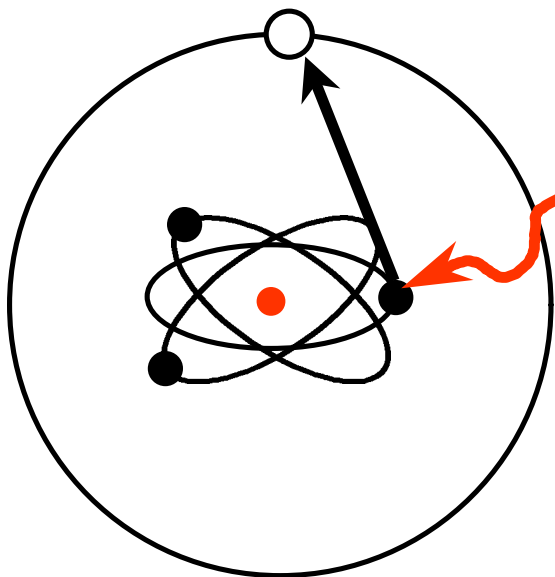


Soft X-ray Physics

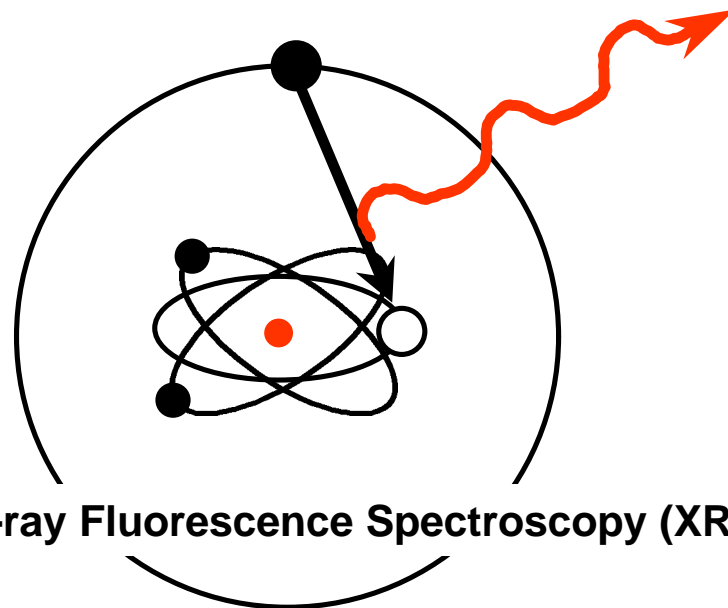
- Overview of research in Prof. Tonner's group
- Introduction to synchrotron radiation physics
- Photoemission spectroscopy: band-mapping and photoelectron diffraction
- Magnetic spectroscopy
- X-ray microscopy and spectro-microscopy

The Three Principle Soft X-ray Spectroscopies

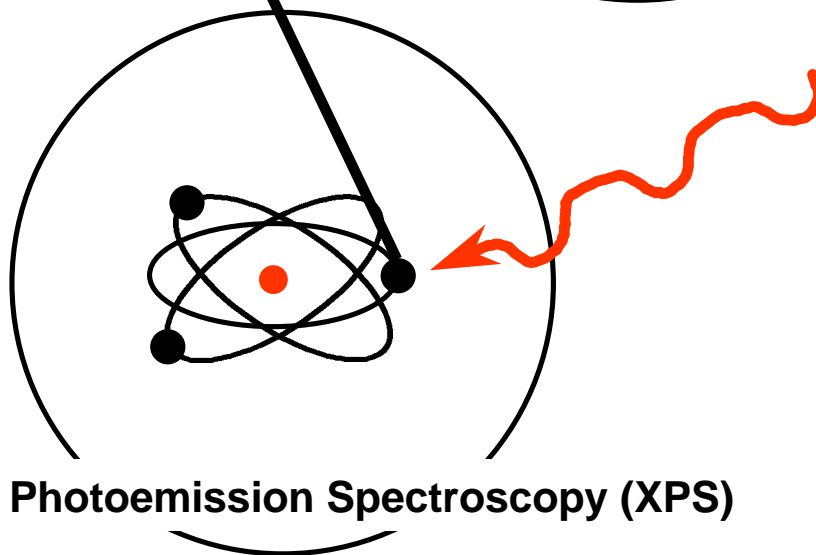
X-ray Absorption Fine Structure (XAFS)



X-ray Fluorescence Spectroscopy (XRF)

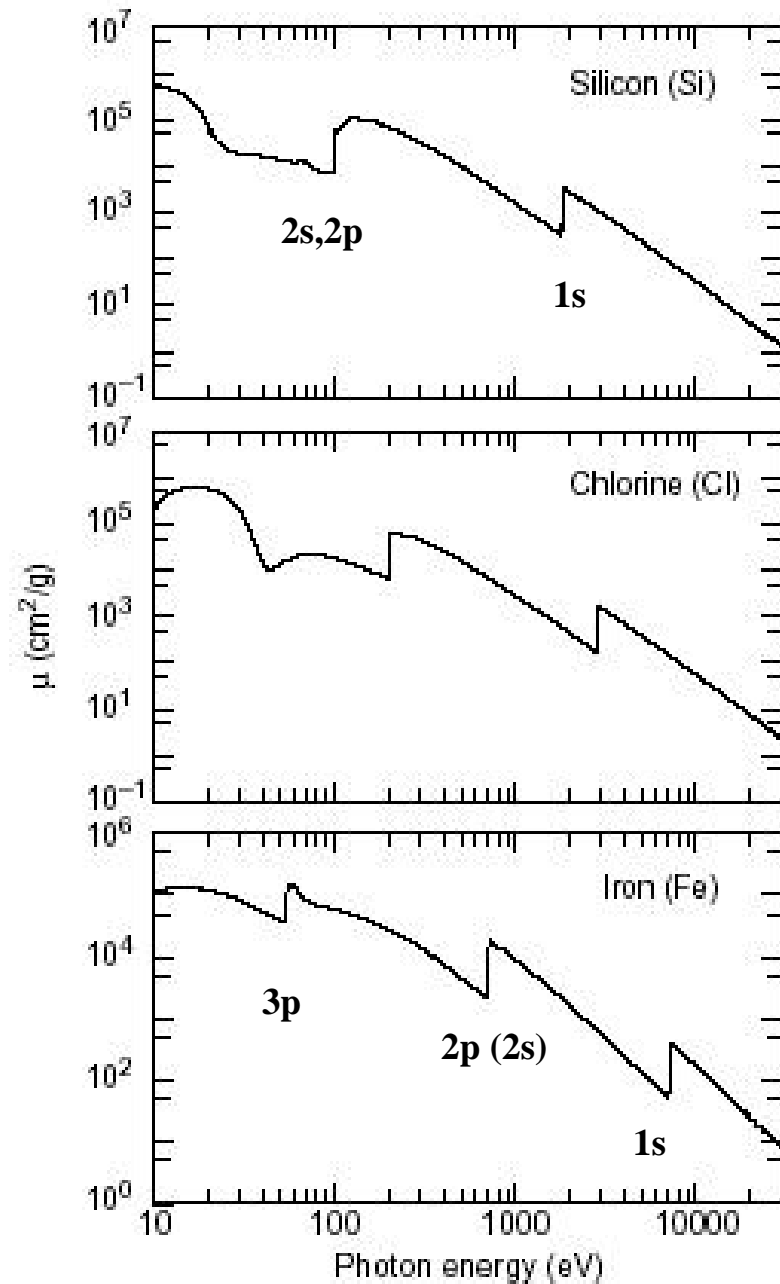


X-ray Photoemission Spectroscopy (XPS)

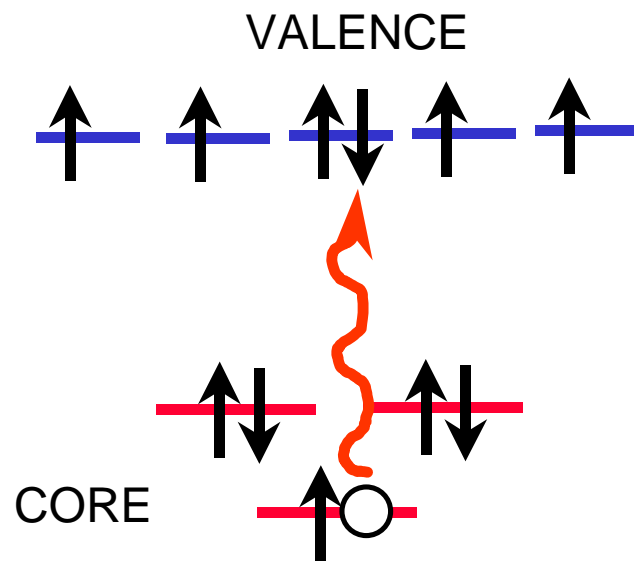
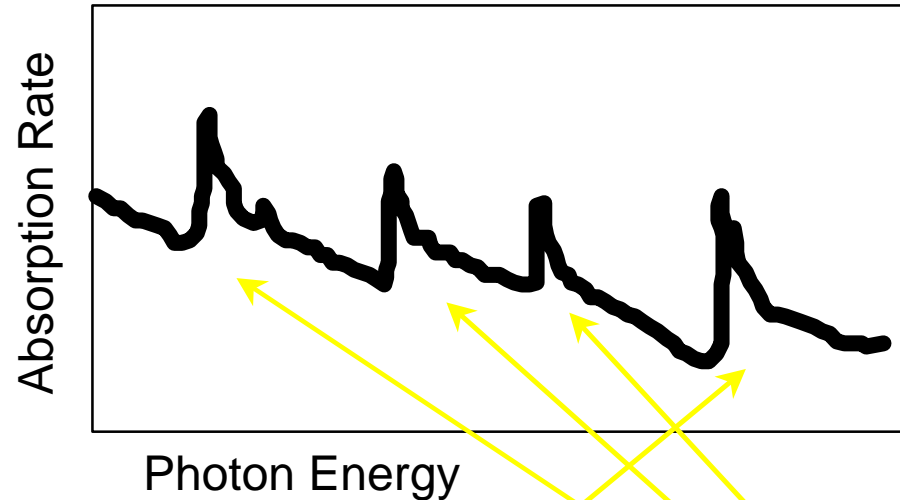
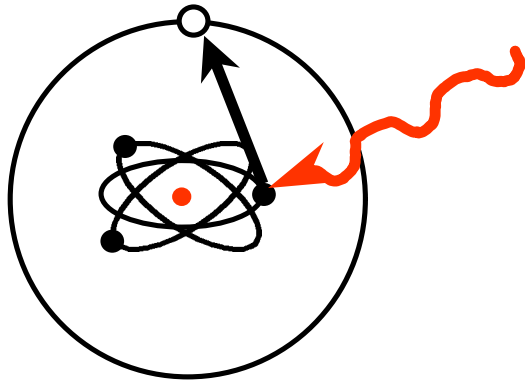


X-ray Absorption Edges

Note the increases in absorption at characteristic energies.



X-ray Absorption Spectroscopy

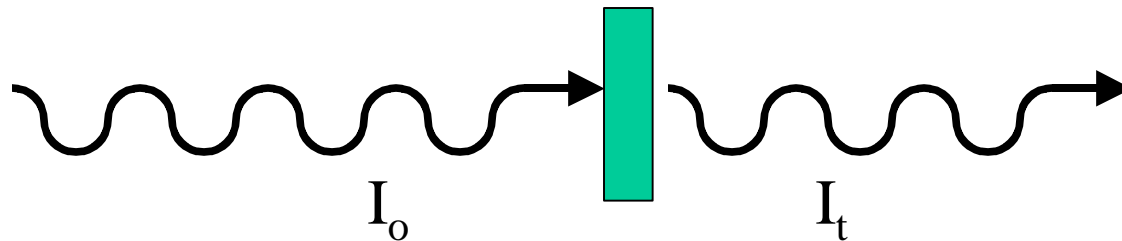


																1																		2									
H		Li		Be																		B		C		N		O		F		Ne											
3		4																		5		6		7		8		9		10													
11		12																		13		14		15		16		17		18													
Na		Mg																		Al		Si		P		S		Cl		Ar													
19		20		21		22		23		24		25		26		27		28		29		30		31		32		33		34		35		36									
K		Ca		Sc		Ti		V		Cr		Mn		Fe		Co		Ni		Cu		Zn		Ga		Ge		As		Se		Br		Kr									
37		38		39		40		41		42		43		44		45		46		47		48		49		50		51		52		53		54									
Rb		Sr		Y		Zr		Nb		Mo		Tc		Ru		Rh		Pd		Ag		Cd		In		Sn		Sb		Te		I		Xe									
55		56		57		72		73		74		75		76		77		78		79		80		81		82		83		84		85		86									
Cs		Ba		La		Hf		Ta		W		Re		Os		Ir		Pt		Au		Hg		Tl		Pb		Bi		Po		At		Rn									
87		88		89		104		105		106		107		108		109		110																									
Fr		Ra		Ac		Rf		Ha		Sg		Ns		Hs		Mt		Uun																									
58		59		60		61		62		63		64		65		66		67		68		69		70		71																	
Ce		Pr		Nd		Pm		Sm		Eu		Gd		Tb		Dy		Ho		Er		Tm		Yb		Lu																	
90		91		92		93		94		95		96		97		98		99		100		101		102		103																	
Th		Pa		U		Np		Pu		Am		Cm		Bk		Cf		Es		Fm		Md		No		Lr																	

Three methods of XAS measurement

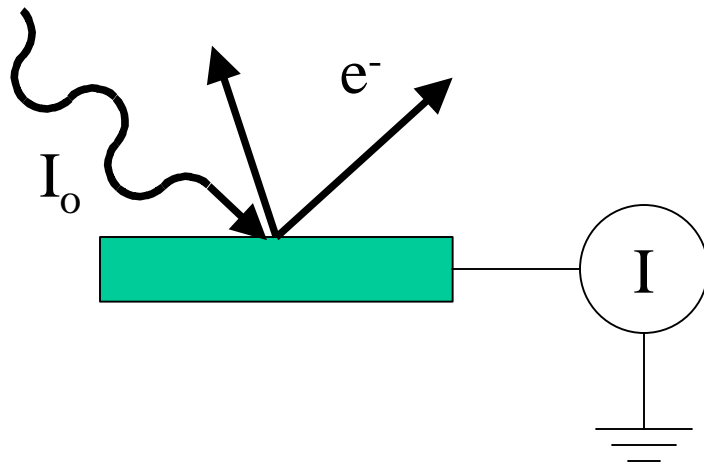
(X-ray Absorption Spectroscopy)

(a) Transmission—bulk properties

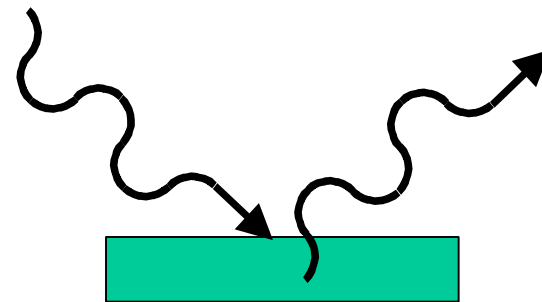


$$I_t = I_0 e^{-m(\hbar\omega)d}$$
$$m(\hbar\omega) = -\frac{1}{d} \ln \left(\frac{I_t}{I_0} \right)$$

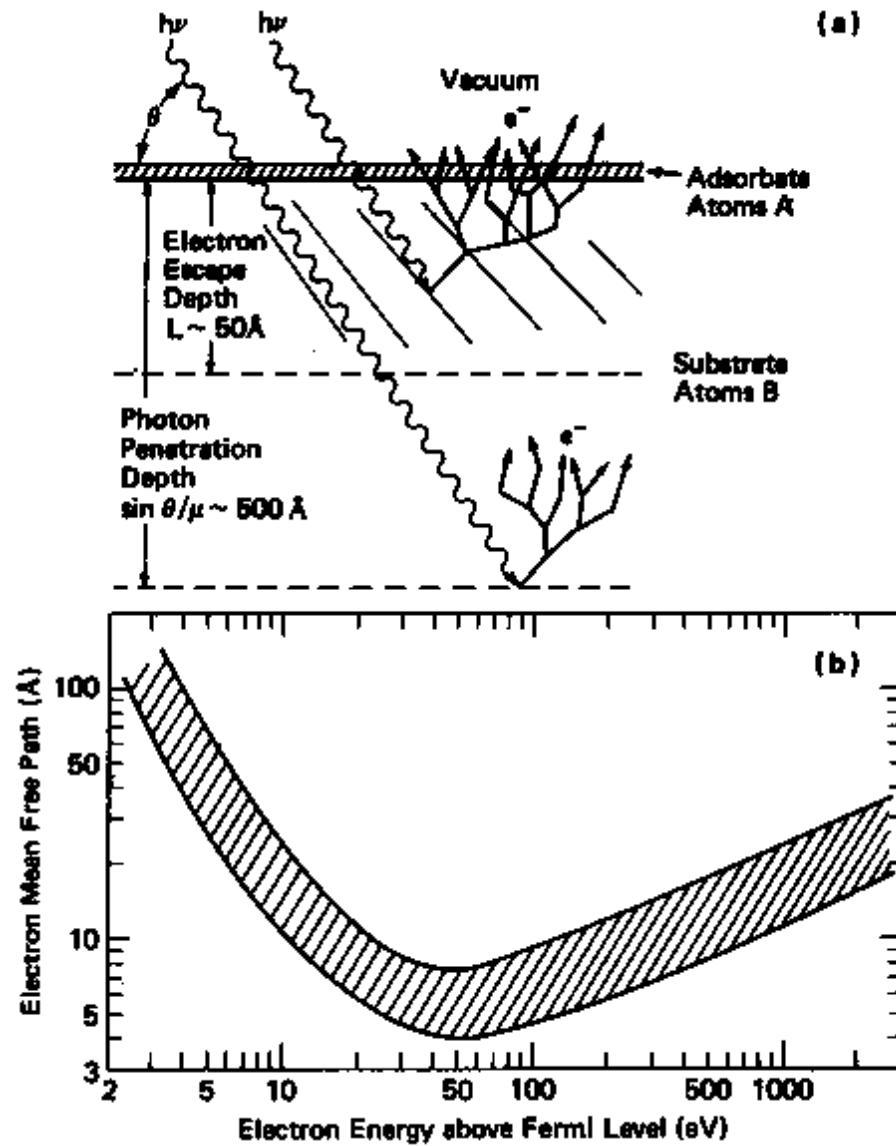
(b) Total electron yield
—surface properties



(c) Fluorescence—dilute species;
Buried interfaces

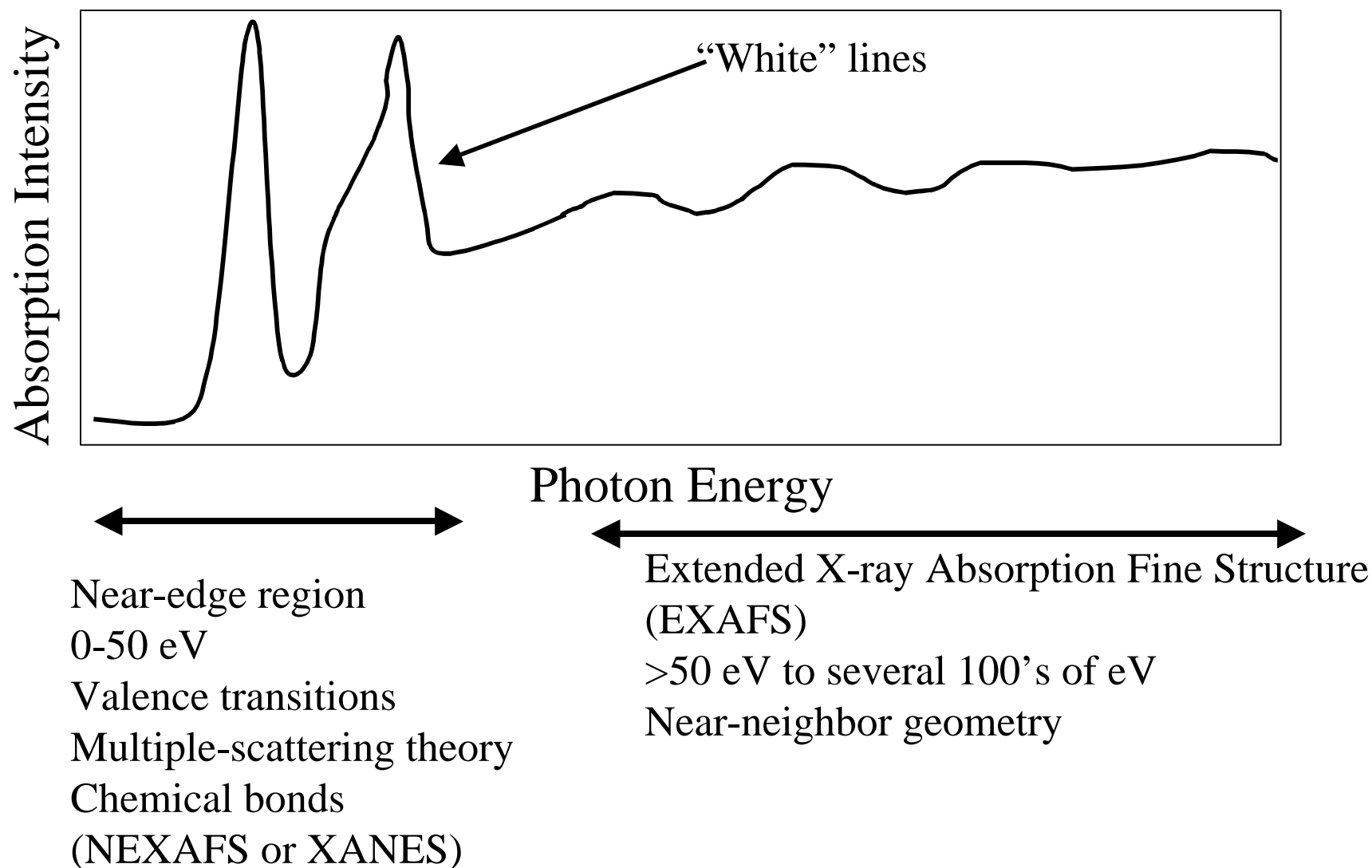


Electron yield X-ray Absorption Spectroscopy

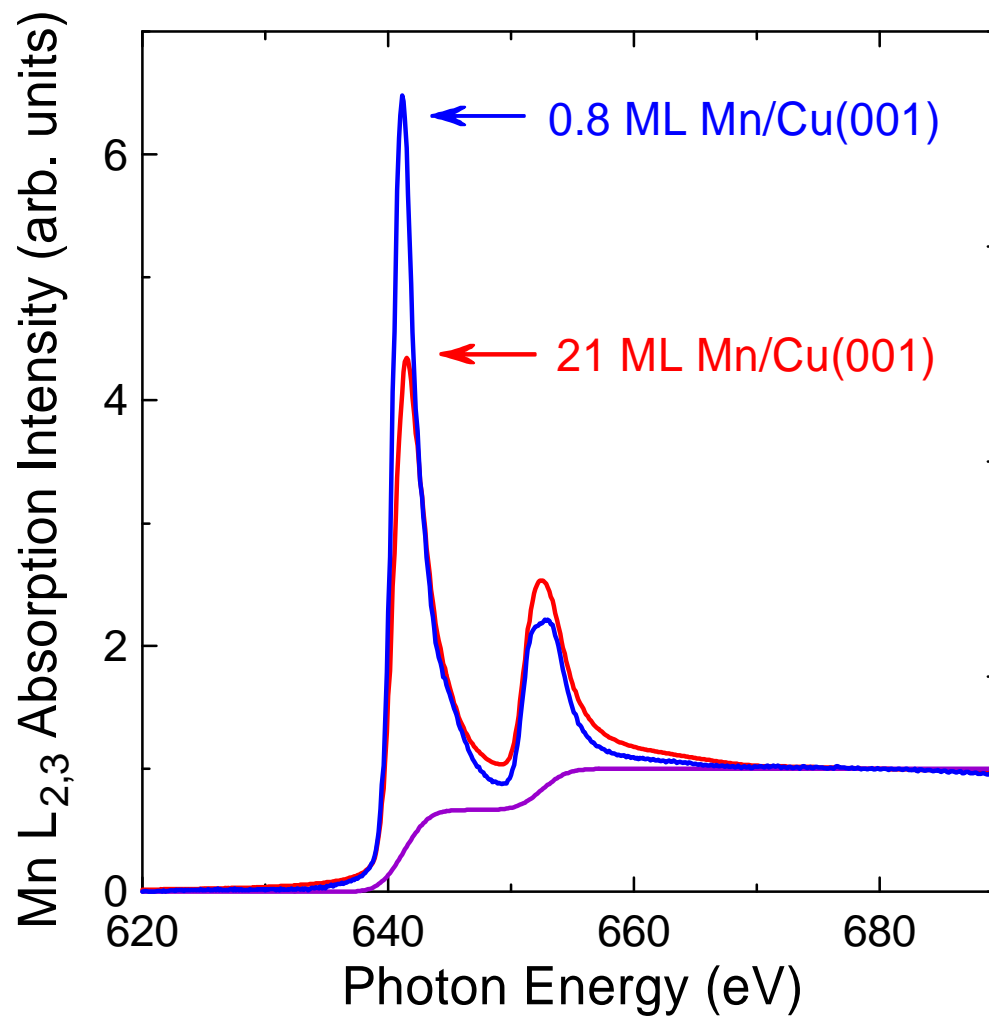


$$i_{\text{yield}} \propto \hbar \omega m(\hbar \omega)$$

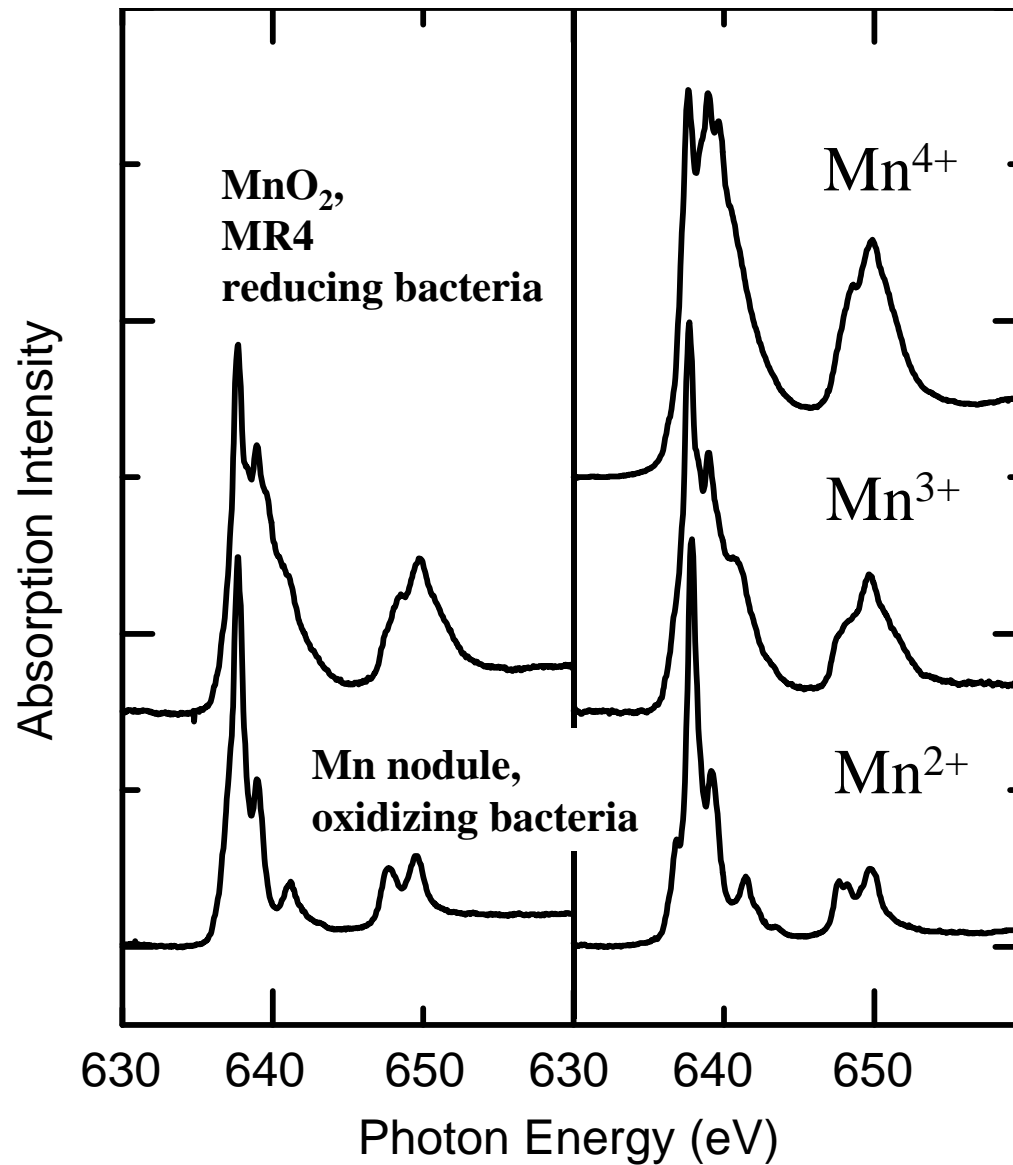
Information in X-ray Absorption Spectroscopy



Background Removal of Continuum Contributions



Mn L-edge XAFS of Bio-inorganic compounds

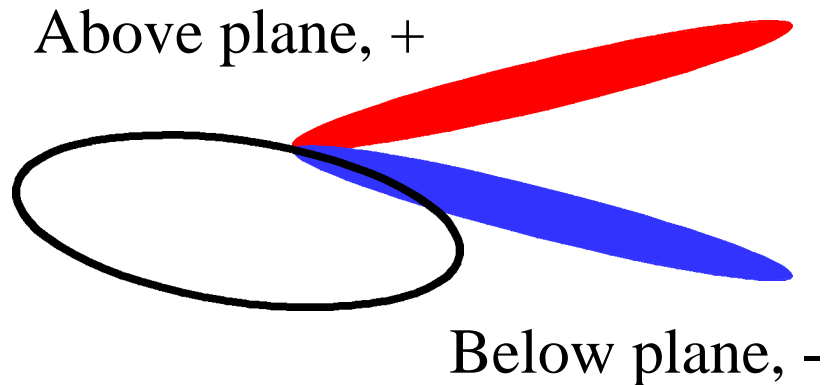


L-edge structure from 2p to 3d dipole allowed transitions

Structure in near-edge can often be explained by atomic theory of crystal field effects

Many materials do not agree with atomic theory, because they have more de-localized orbitals—need a many-body theory

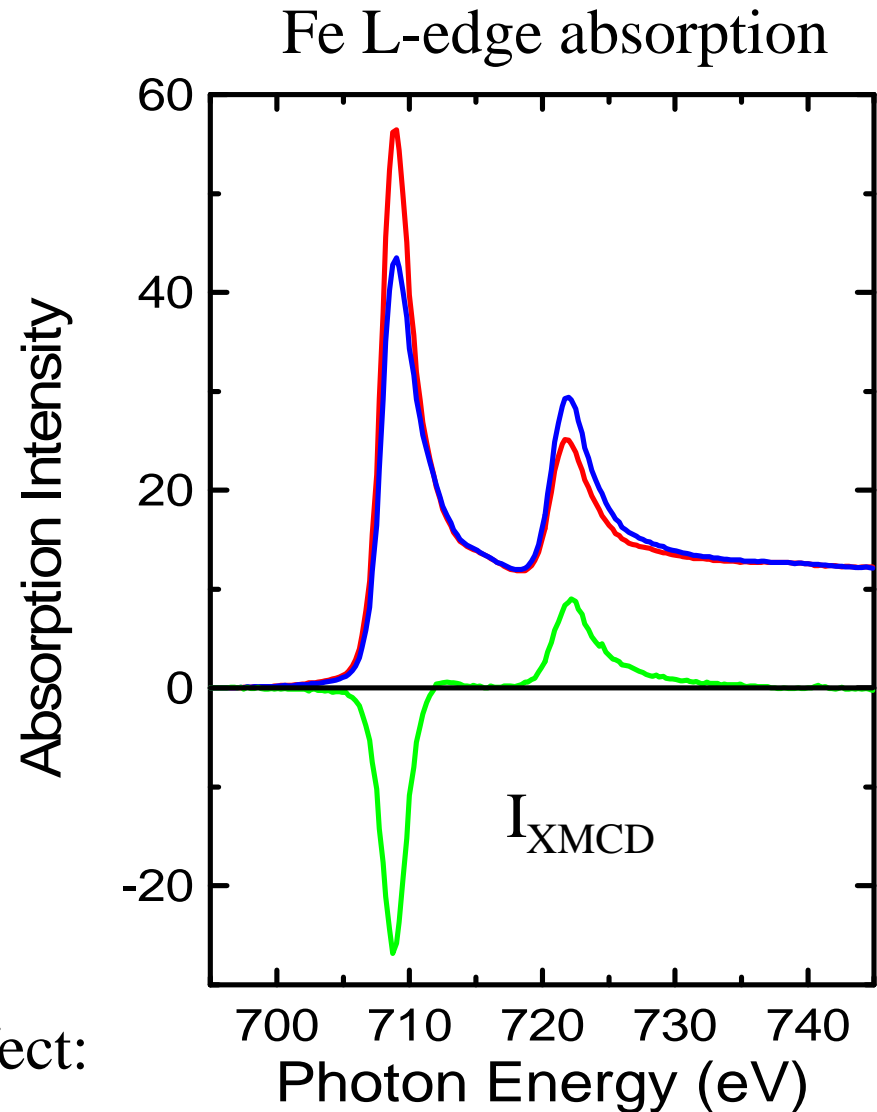
XAS with Circular Polarization



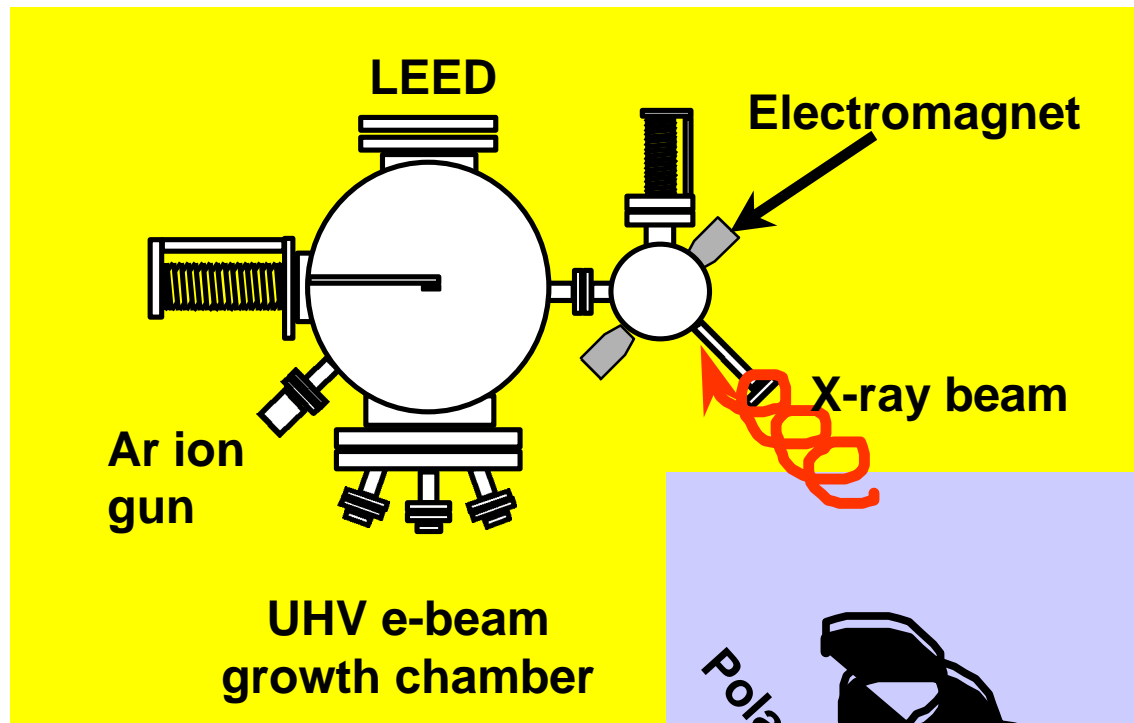
Photons carry angular momentum (spin), which is parallel or anti-parallel to the direction of propagation for circularly polarized light.

$$I_{XMCD} \propto \vec{\Sigma} \cdot \vec{M}$$

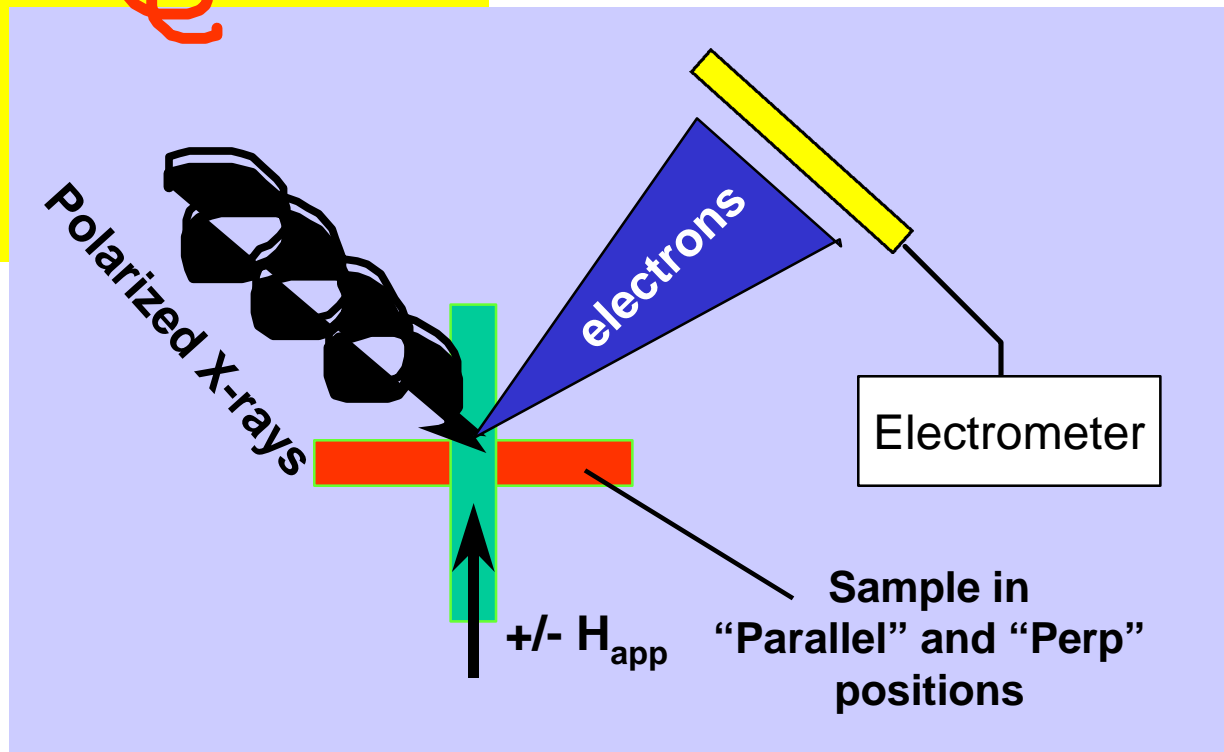
The effect is an atomic physics effect:
Spin-orbit splitting of d-shell electrons.



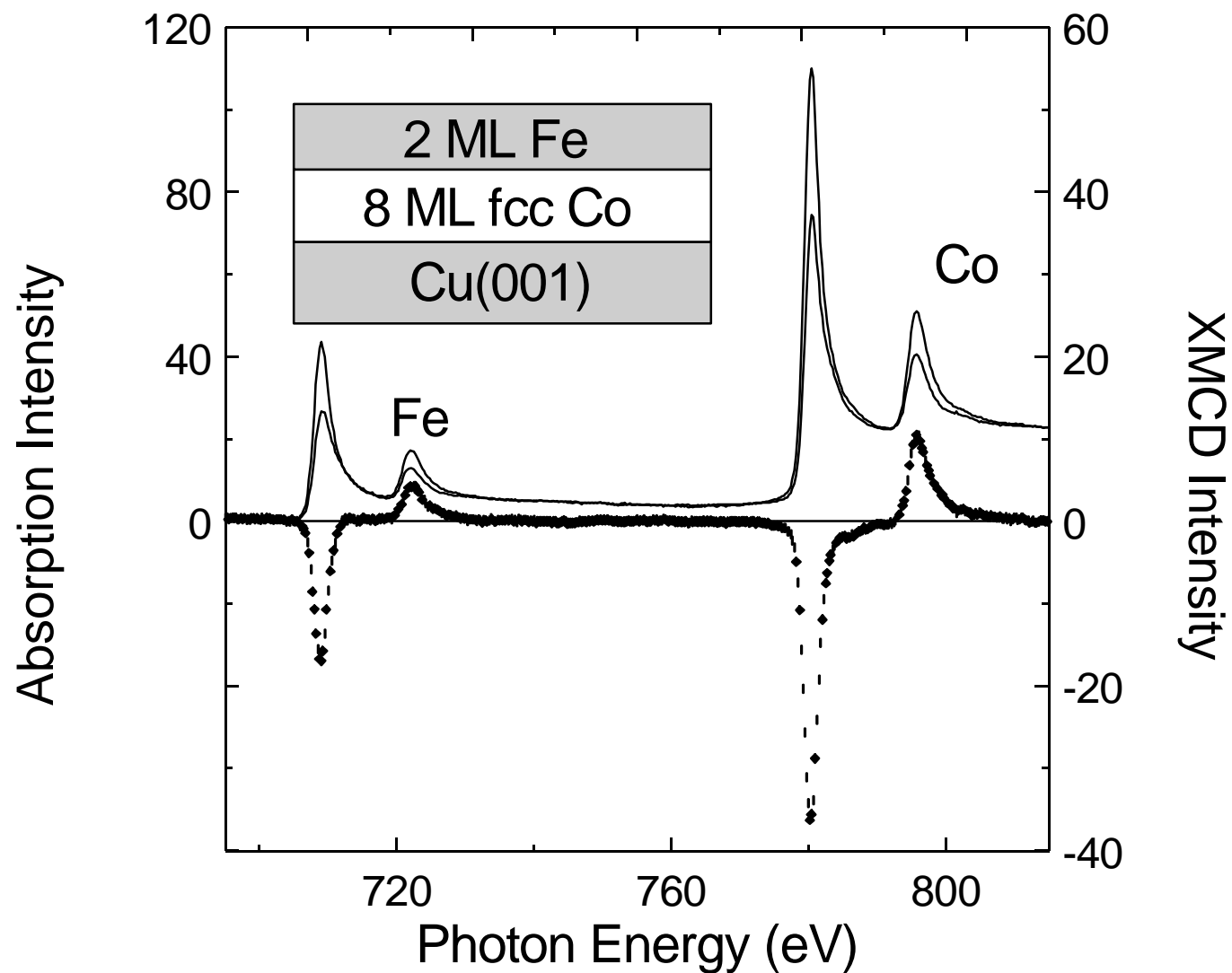
The technique of X-ray Magnetic Circular Dichroism (XMCD)



$$I_{XMCD} \propto \vec{\Sigma} \cdot \vec{M}$$

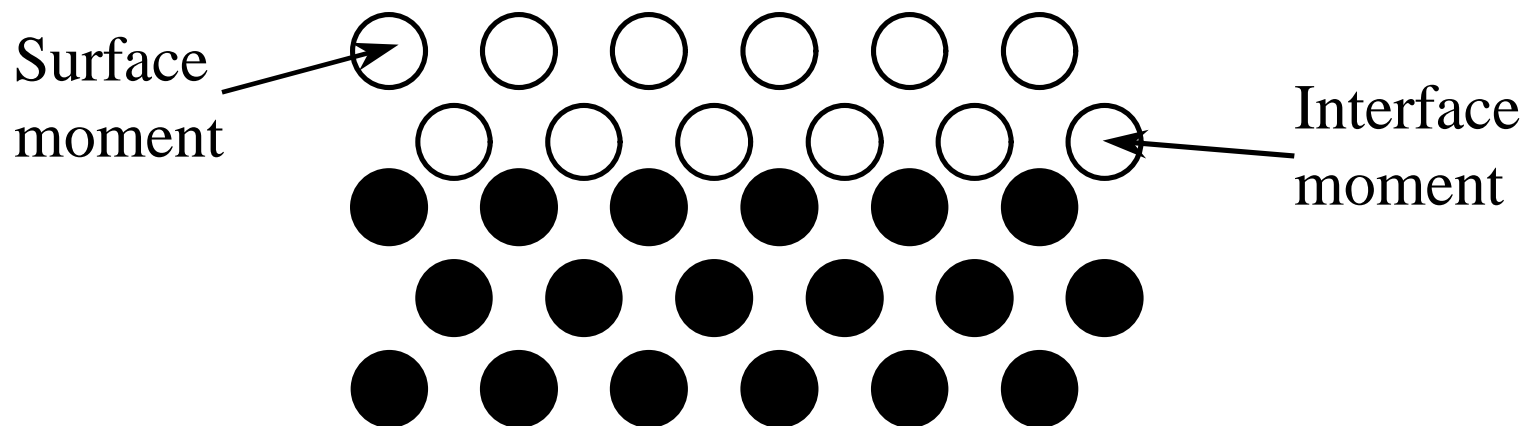


XMCD: Element specific magnetometry



Spin and Orbital Contributions to Magnetic moment

- Total moment is $\mathbf{M}=(\mathbf{L} + 2\mathbf{S})\mu_B$.
- For (Fe, Co, Ni), $L/2S$ is about 1/10, so the orbital moment is small, but....
- Orbital moments contribute significantly to the magnetic anisotropy (spin-orbit)



Orbital moments may be enhanced at surfaces or interfaces.

Dichroism sum rules

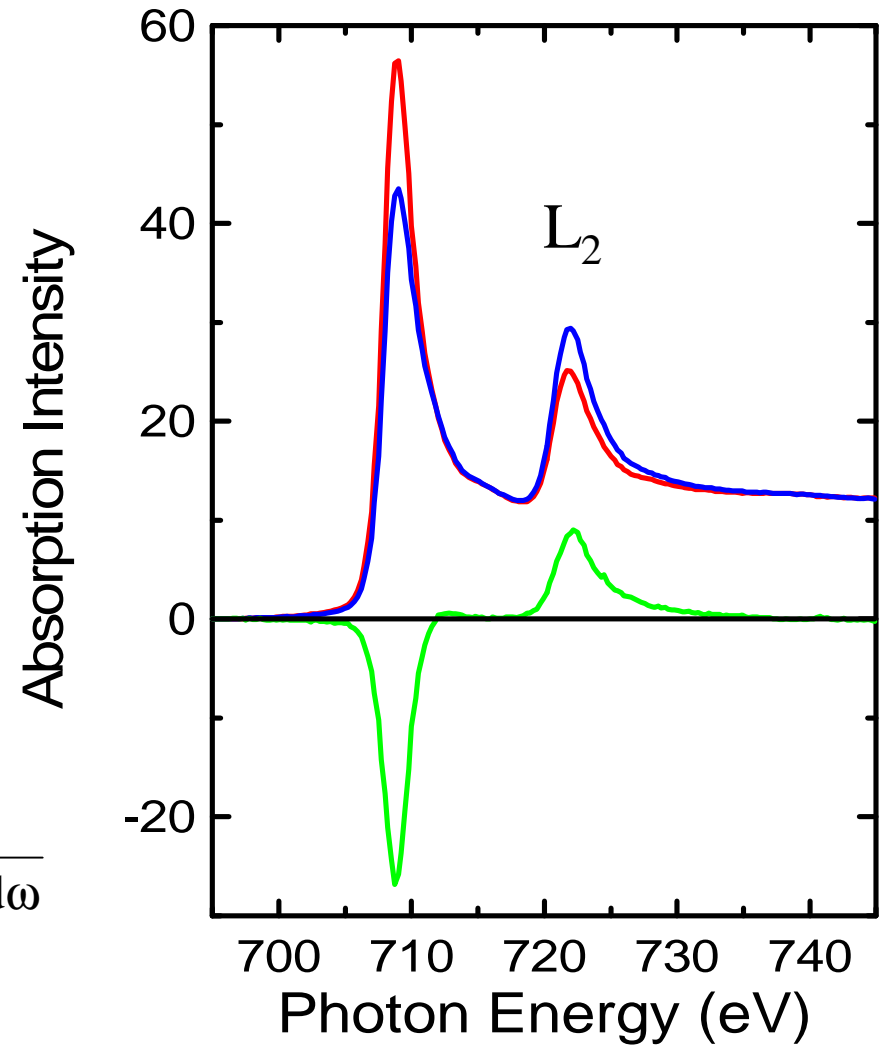
$$M_{\text{orbit}} = L_3 + L_2$$

$$M_{\text{spin}} = |L_3| + 2|L_2| \quad L_3$$

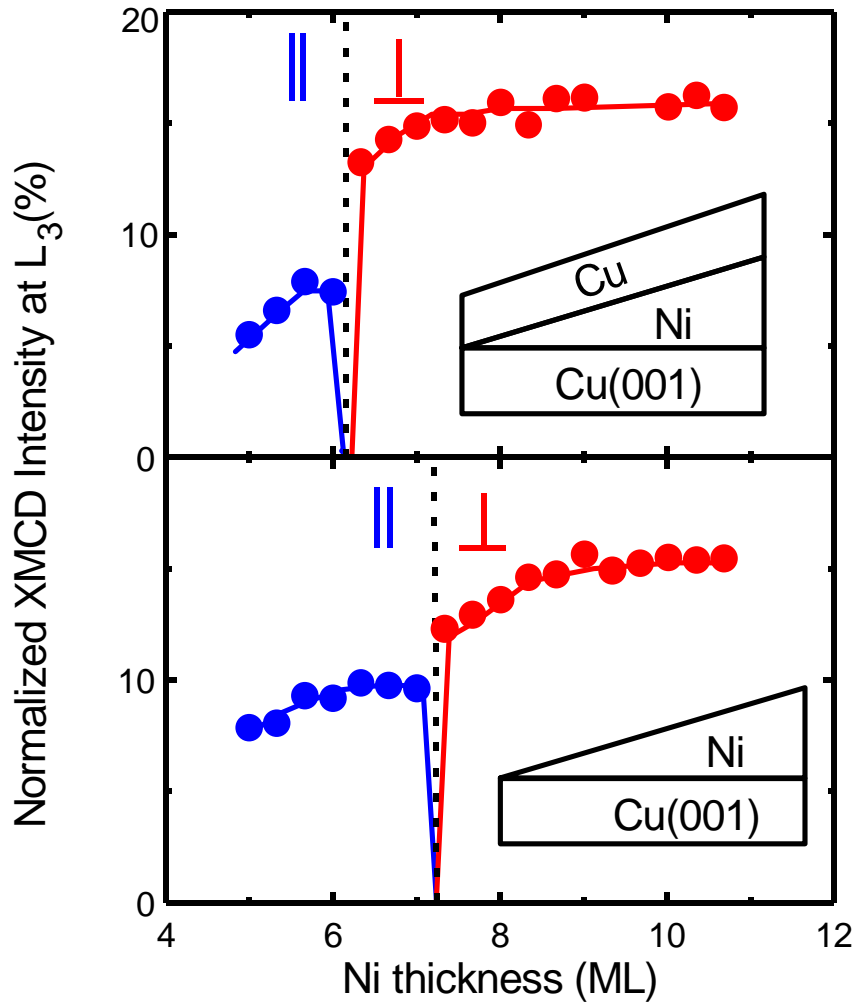
$$M_o = \frac{4}{3}h \frac{\int \sigma_M d\omega}{\int (\sigma_+ + \sigma_-) d\omega} \quad L_{2,3}$$

$$M_S + 7M_D = 2h \frac{L_3}{L_2} \frac{\int \sigma_M d\omega - 2 \int \sigma_M d\omega}{\int (\sigma_+ + \sigma_-) d\omega} \quad L_{2,3}$$

$$\frac{M_o}{M_S + 7M_D} = \frac{2}{3} \cdot \frac{\int \sigma_M d\omega}{\int \sigma_M d\omega - 2 \int \sigma_M d\omega} \quad L_{2,3} \quad L_2$$



XMCD of Magnetic properties of ultrathin films



Schulz and Baberschke* have already determined $K_I + K_S$ (Interface plus surface anisotropy) to be -0.38 ergs/cm^2

Note this favor *in-plane* M

Using this value and the results presented here we determine that

$$K_I = -0.16 \text{ ergs/cm}^2$$

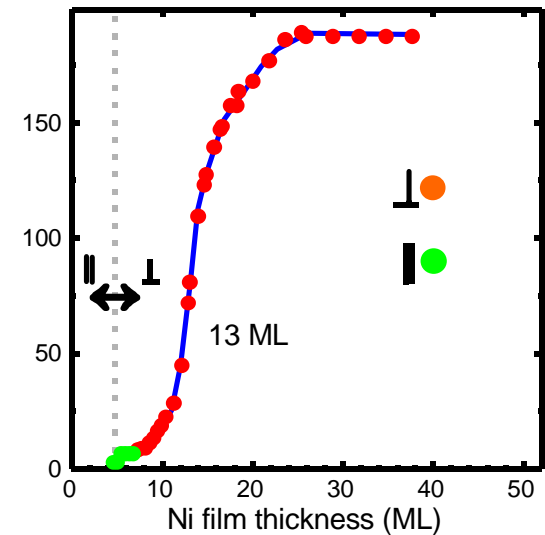
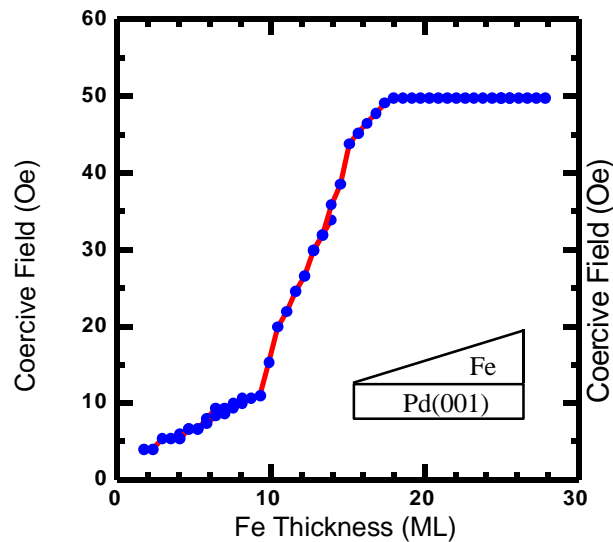
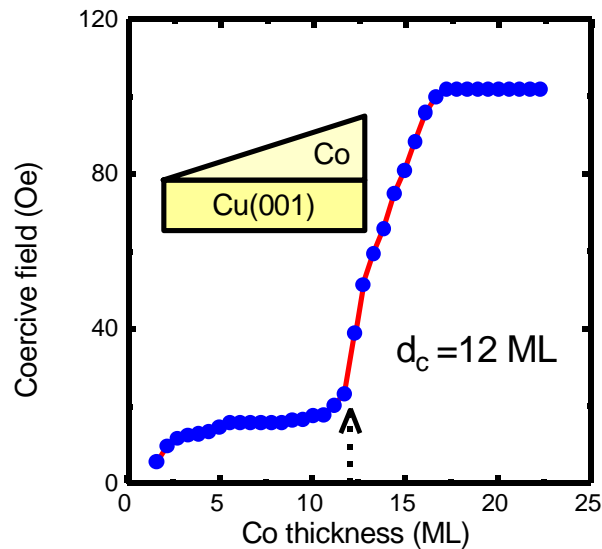
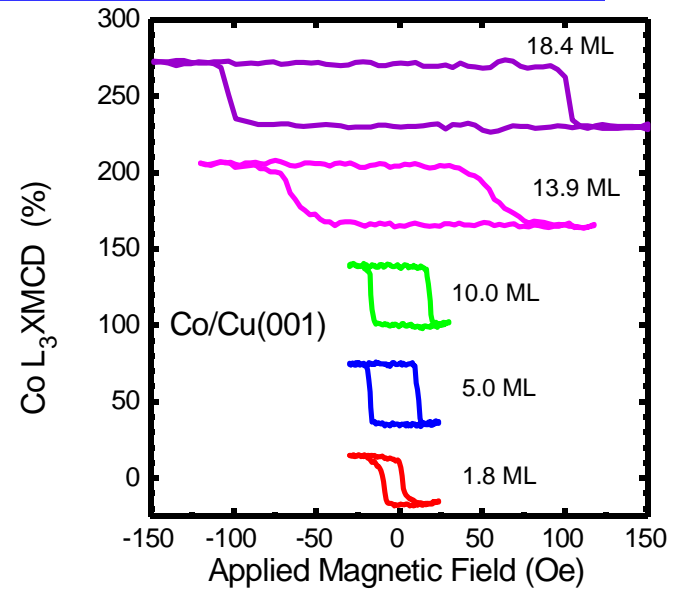
$$K_S = -0.22 \text{ ergs/cm}^2$$

This means: both interface and surface anisotropy are negative.

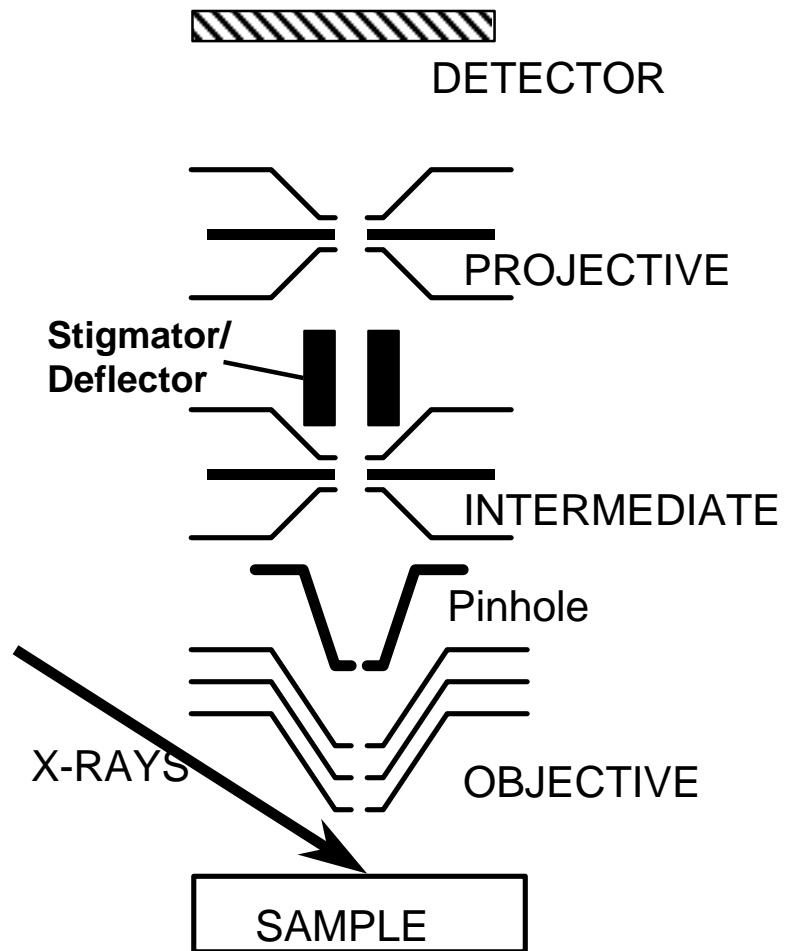
*PRB **50** 13468 (1994).

XMCD Magnetometry of Ultrathin films

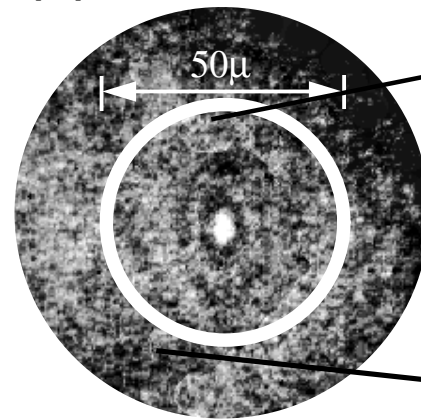
- Main features are film independent
- Coercivities rise sharply near the critical thickness



XMCD Microscopy

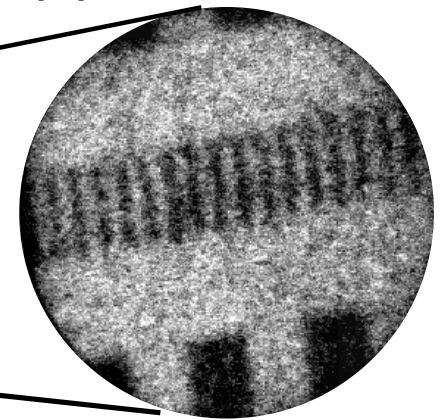


(A)



Linear polarization:
Topographical information

(B)

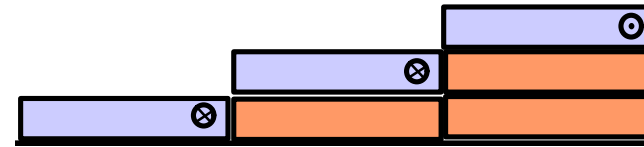
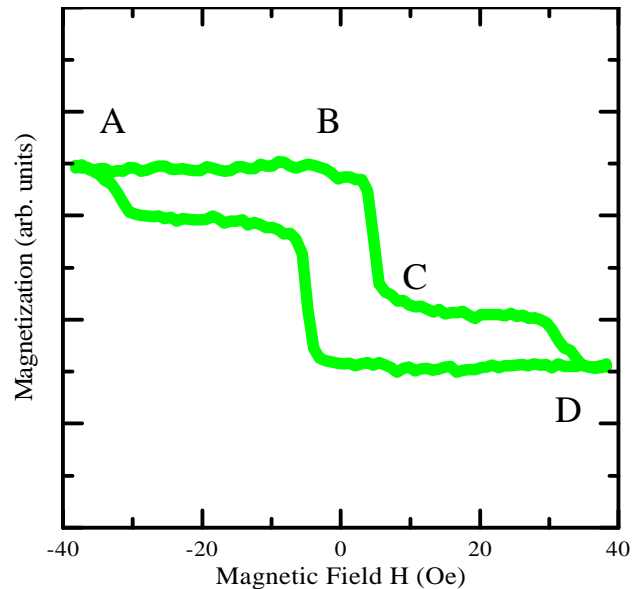


Circular polarization
difference image:
Magnetic bit information

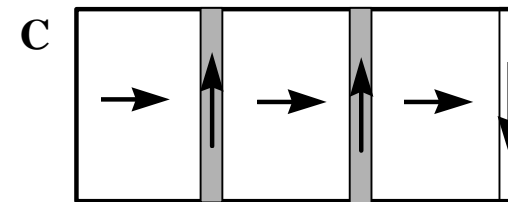
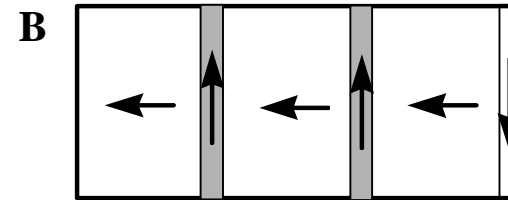
Step bunches can explain anomalous uniaxial anisotropy

Two different sites for atoms

- near steps - strong uniaxial anisotropy
- terrace - weaker uniaxial anisotropy (non-zero)



A - fully magnetized ←

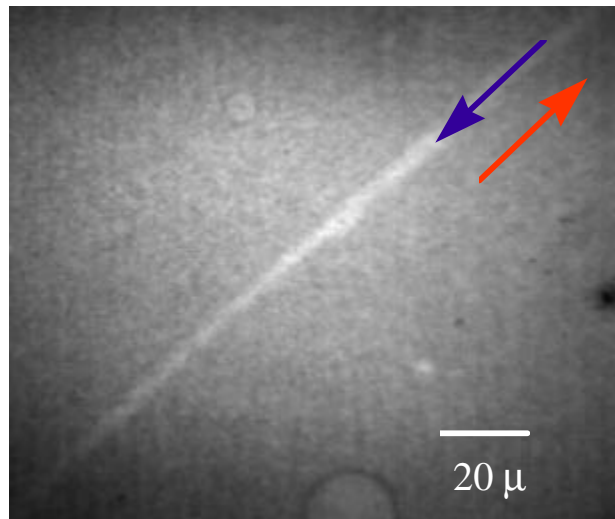


D - rotate the step moments
full magnetization

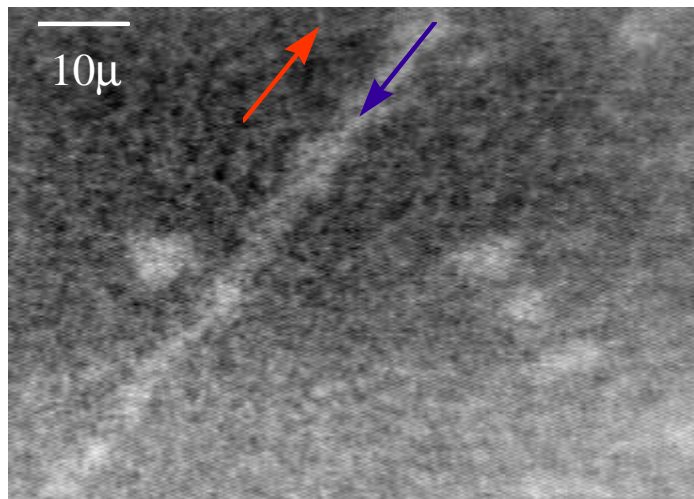
Magnetization Reversal along anomalous axis: Schematic explaining the magnetization reversal along the anomalous “easy” axis of magnetization for miscut fcc Co films. There are two spin sites, a terrace site which has a weak uniaxial anisotropy which may or may not be zero and a step site which has a strong uniaxial anisotropy.

XMCD microscopy of step bunch domains

Co/Cu ultrathin films



Spontaneous domain formation



Hard-axis magnetization

110 Easy axis

